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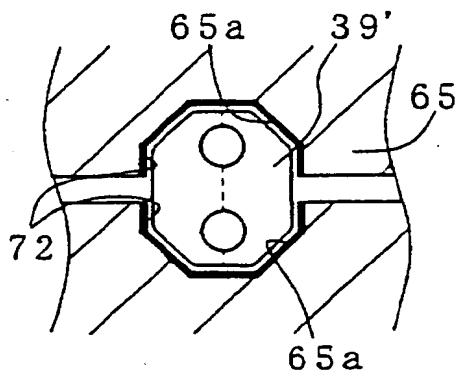
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(54) Ceramic heater and method for manufacturing the same

(57) A ceramic heater includes a silicon nitride ceramic substrate (13) and a heating member (10) embedded in the silicon nitride ceramic substrate (13). An Al-augmented layer (2s) is formed in a surface layer portion of the silicon nitride ceramic substrate (13). The Al-aug-

mented layer (2s) has an Al concentration higher than that of an internal portion of the silicon nitride ceramic substrate (13). The Al-augmented layer (2s) assumes such a graded composition structure that the Al concentration increases toward the surface of the silicon nitride ceramic substrate (13).

FIG. 10A



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FIG. 10B

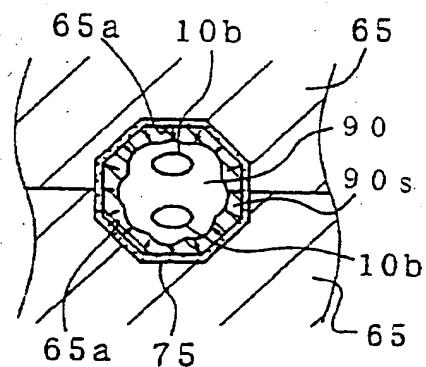


FIG. 10C

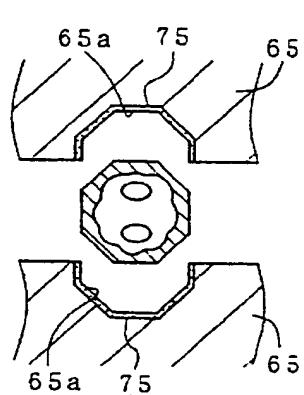
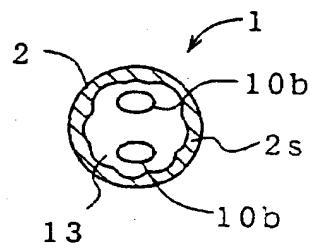


FIG. 10D



Description

[0001] The present invention relates to a ceramic heater for use in, for example, a glow plug, as well as to a method for manufacturing the ceramic heater.

5 [0002] A conventional ceramic heater for use in, for example, a ceramic glow plug includes an insulating ceramic substrate and a resistance-heating member embedded in the ceramic substrate and formed of, for example, a conductive ceramic material. Because of excellent thermal shock resistance and high-temperature strength, silicon nitride ceramic is a popular material for the ceramic substrate.

10 [0003] As in the case of manufacture of many other ceramics, manufacture of silicon nitride ceramic employs a sintering aid. A sintering aid melts into a liquid phase during firing to thereby accelerate densification of a ceramic sintered body to be obtained, and plays a role in forming a grain boundary phase for bonding silicon nitride main phase (grains) of the ceramic sintered body. Sintering aids which are commonly used in manufacturing silicon nitride ceramic include magnesium oxide (MgO) and a combination of alumina (Al_2O_3) and yttria (Y_2O_3). However, these sintering aids involve a drawback in that a vitric grain boundary phase, whose softening point is low, tends to be formed in firing, and thus high-temperature strength, particularly strength at 1200°C or higher, of an obtained sintered body tends to be impaired. When a rare-earth metal oxide and silica are added as sintering aids, the grain boundary phase can be crystallized to thereby contribute to enhancement of high-temperature strength of a sintered body. However, since a liquid phase generated during firing exhibits poor fluidity, the sintering speed lowers. As a result, nonuniform sintering tends to occur, and thus variations in room-temperature strength tend to occur among sintered bodies.

20 [0004] In order to cope with the above problems, Japanese Patent No. 2735721 discloses a silicon nitride ceramic heater whose manufacture employs a rare-earth metal oxide and alumina as sintering aids. Conceivably, addition of Al_2O_3 appropriately improves the sintering speed and increases the strength of the grain boundary phase.

25 [0005] However, in the technique disclosed in the above publication, since Al_2O_3 is added uniformly to the entirety of the ceramic sintered body, the softening point of the grain boundary phase tends to lower. Thus, high-temperature strength unavoidably lowers. Also, since an Al component is contained in the form of Al_2O_3 , the grain boundary phase tends to be vitrified, causing an adverse effect on the intended improvement of high-temperature strength.

30 [0006] An object of the present invention is to provide a ceramic heater whose silicon nitride ceramic substrate substantially maintains high-temperature strength without impairment resulting from addition of an Al component, and featuring low variations in room-temperature strength among silicon nitride ceramic substrates.

[0007] Another object of the present invention is to provide a method for manufacturing the above-described ceramic heater.

35 [0008] To achieve the above object, the present invention provides a ceramic heater comprising a silicon nitride ceramic substrate (hereinafter, may be called merely a ceramic substrate or a substrate) and a resistance-heating member embedded in the silicon nitride ceramic substrate. An Al-augmented layer is formed in a surface layer portion of the silicon nitride ceramic substrate. The Al-augmented layer has an Al concentration higher than that of an internal portion of the silicon nitride ceramic substrate.

40 [0009] According to the above configuration, the Al-augmented layer is formed merely in a surface layer portion of the ceramic substrate. Thus, even when the Al component causes lowering of the softening point of the grain boundary phase, the influence is limited to the surface layer portion of the ceramic substrate. Therefore, the high-temperature strength of the ceramic substrate is unlikely to be impaired. Formation of the Al-augmented layer suppresses growth of silicon nitride main phase (hereinafter, may be called merely main phase) grains in the surface layer portion of the ceramic substrate, so that abnormally grown grains which provide a starting point of fracture are hardly produced. Therefore, the Al-augmented layer prevents variations in strength, particularly room-temperature strength, among ceramic substrates.

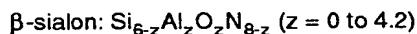
45 [0010] Preferably, the thickness of the Al-augmented layer is 50 μm to 1000 μm . When the thickness is less than 50 μm , variations in room-temperature strength among silicon nitride ceramic substrates may fail to be effectively suppressed. When the thickness is in excess of 1000 μm , sufficient high-temperature strength may fail to be imparted to the ceramic substrate. More preferably, the thickness is 50 μm to 500 μm . Preferably, the Al-augmented layer assumes an average Al concentration of 0.1% to 5% by weight. When the Al concentration is less than 1% by weight, growth of 50 main phase grains in the surface layer portion of the ceramic substrate may fail to be effectively suppressed, potentially causing variations in room-temperature strength among ceramic substrates. When the Al concentration is in excess of 5% by weight, the high-temperature strength of the Al-augmented layer itself may be impaired, potentially failing to attain enhancement of strength of the ceramic substrate intended by means of the Al-augmented layer.

55 [0011] When growth of main phase grains in the Al-augmented layer is effectively suppressed, the average grain size of the silicon nitride main phase in the Al-augmented layer becomes smaller than that of the silicon nitride main phase in an internal portion of the ceramic substrate. Preferably, the average grain size of the silicon nitride main phase in the Al-augmented layer is 0.1 μm to 1 μm , and that of the silicon nitride main phase in the internal portion is 0.2 μm to 5 μm . In either case, when the average grain size is below the lower limit, preparation of material powder for attain-

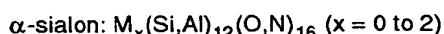
ment of the average grain size becomes very difficult. When the average grain size is in excess of the upper limit, the strength of the ceramic substrate may become insufficient. In order to suppress formation of a starting point of fracture, the maximum grain size of the main phase in the Al-augmented layer is preferably not greater than 10 µm. Herein, the grain size is defined as follows. Various pairs of parallel lines are drawn tangentially to the contour of a crystal grain observed on a sectional microstructure of the ceramic substrate, in such a manner as not to traverse the crystal grain. The distance between the parallel lines of each pair is measured. The maximum distance is defined as the grain size of the crystal grain.

[0012] The silicon nitride ceramic substrate assumes, for example, a microstructure such that Si_3N_4 grains are bonded by means of a grain boundary phase (bonding phase) derived from a sintering aid component, which will be described later. Preferably, the main phase is predominantly composed of an Si_3N_4 phase which contains $\beta\text{-}\text{Si}_3\text{N}_4$ in an amount of not less than 70% by volume (preferably, not less than 90% by volume). In this case, the Si_3N_4 phase may be such that a portion of Si or N atoms may be replaced with Al or oxygen atoms and such that metallic atoms of, for example, Li, Ca, Mg, or Y, are incorporated into the phase in the form of solid solution. Examples of such a phase construction are sialon expressed by the following formulas.

15



20



where M represents Li, Mg, Ca, Y, or R (a rare-earth element other than La and Ce). Herein, unless otherwise specified, the term "predominant" or "predominantly" used in relation to content means that the content of a substance in question is contained in an amount of not less than 50% by weight.

[0013] Preferably, a predominate portion of the Al component in the ceramic substrate is present in the form of an inorganic compound other than Al_2O_3 . Specifically, it is preferable that the Al component be integrated into the main phase through replacing a silicon component as mentioned above and that the Al component present in the grain boundary phase assume the form of nitride or oxynitride or the form of composite nitride, composite oxide, or composite oxynitride with another sintering aid component. The Al component present in such a form suppresses vitrification of the grain boundary phase or further enhances suppression of growth of crystal grains in the main phase, thereby effectively suppressing impairment in the high-temperature strength of the ceramic substrate. This effect is developed irrespective of whether or not the surface layer portion of the ceramic substrate is augmented with the Al component.

[0014] The present invention further provides a ceramic heater comprising a silicon nitride ceramic substrate containing an Al component and a resistance-heating member embedded in the silicon nitride ceramic substrate. The Al component is present in the silicon nitride ceramic substrate predominantly in the form of an inorganic compound other than Al_2O_3 .

[0015] Next, a sintering aid component is predominantly engaged in formation of the bonding phase; however, a portion of the sintering aid component may be integrated into the main phase. The bonding phase may contain unavoidable impurities; for example, silicon oxide contained in silicon nitride material powder, in addition to an intentionally added component serving as a sintering aid.

[0016] A sintering aid component usable in the present invention is not limited to a rare-earth component. For example, elements of Groups 4A, 5A, 3B, and 4B of the Periodic Table, such as Si and Al, can be used to such an extent as not to impair the effect of the present invention. The silicon nitride ceramic substrate to be obtained may contain a sintering aid component in an amount of 3% to 15% by weight on an element basis. When the content of the sintering aid component is less than 3% by weight, a dense sintered body is hardly obtained. When the content of the sintering aid component is in excess of 15% by weight, strength, toughness, or heat resistance may become insufficient. The content of the sintering aid component is preferably 6% to 10% by weight.

[0017] Rare-earth components usable in the present invention are Sc, Y, La, Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, Yb, and Lu. In particular, Tb, Dy, Ho, Er, Tm, and Yb can be favorably used, since the elements, when added, accelerate crystallization of the grain boundary phase and improve high-temperature strength. Combined addition of a rare-earth component and the Al component synergistically suppresses abnormal growth of crystal grains in the main phase and variations in room-temperature strength among ceramic substrates (in particular, the synergistic suppression becomes more effective when one or more rare-earth elements among Tb, Dy, Ho, Er, Tm, and Yb are used). Conceivably, this is because presence of the above rare-earth element(s) facilitates presence of the Al component in the form of an inorganic compound other than Al_2O_3 . The synergistic effect is maximally yielded when the Al content of the ceramic substrate is 0.1% to 5% by weight, and the rare-earth component content of the ceramic substrate is 3% to 15% by weight.

[0018] In order to accelerate crystallization of the grain boundary phase, it is preferable that a rare-earth component

serving as a sintering aid component be present in the form of a composite oxide with Si; specifically, the rare-earth component be contained in the form of R_2SiO_5 and/or $R_2Si_2O_7$, where R is the rare-earth element. Whether or not the composite oxides are contained in a crystalline form can be confirmed by various known processes of crystal analysis; for example, X-ray diffractometry for measuring diffraction patterns, or selected-area diffractometry using a transmission electron microscope.

[0019] In order to accelerate crystallization of the grain boundary phase, an excess silicon oxide component which is not involved in formation of the main phase must be contained in a necessary and sufficient amount. The excess silicon oxide component accelerates, for example, formation of crystals of the aforementioned composite oxide. The content of the excess silicon oxide component can be estimated from the amount of excess oxygen, which will be described below. The amount of sintering aid components, excluding Si and Al, contained in the ceramic substrate and the total oxygen content of the ceramic substrate are obtained. The amount of oxygen required for all the sintering aid components to be present in the form of oxide is subtracted from the total oxygen content. The thus-obtained amount of oxygen is defined as the amount of excess oxygen. In order to reliably yield the aforementioned effect through addition of a rare-earth component as a sintering aid component, the amount of excess oxygen is preferably 1% to 10% by weight on an SiO_2 basis. When the amount of excess oxygen is less than 1% by weight on the SiO_2 basis, sinterability is impaired. When the amount of excess oxygen is in excess of 10% by weight, the softening point of the grain boundary phase drops, potentially resulting in impairment in high-temperature strength.

[0020] Preferably, the Al-augmented layer assumes such a graded composition structure that the Al concentration increases toward the surface of the ceramic substrate. Through employment of the structure, a compositionally discontinuous boundary portion is unlikely to be formed between the Al-augmented layer and the internal portion of the ceramic substrate, thereby further enhancing the strength of the ceramic substrate.

[0021] When a gradient is present in the Al concentration of the Al-augmented layer, the thickness t of the Al-augmented layer is defined as shown in FIG. 11. Specifically, an Al concentration curve is obtained through measurement in the direction of depth x from the surface of the silicon nitride ceramic substrate. The Al concentration curve is expressed by

$$C = F(x) \quad (1)$$

[0022] On the curve (1), C_0 is an Al concentration by weight measured on the surface of the substrate, and CB is an average Al concentration by weight of an internal portion of the substrate (CB may be substantially zero in some cases). A straight line is defined as follows.

$$C = CB + 0.5(C_0 - CB) \quad (2)$$

[0023] On the x-C plane, the intersection B of the curve (1) and the straight line (2) is obtained (when a plurality of intersections are formed due to influence of noise appearing on the curve (1), the one closest to the substrate surface is selected). The x coordinate of the intersection B is defined as the thickness t of the Al-augmented layer. The Al concentration curve can be obtained by a so-called line analysis. In this line analysis, an analysis line AL is first defined on the cross section of the substrate in the above-mentioned x direction, and the concentration of Al is determined along the analysis line AL by making use of an electron probe X-ray microanalyzer (EPMA). The Al concentration curve is obtained as a profile of variation in characteristic X-ray intensity of the Al component along the analysis line AL. Herein, the characteristic X-ray intensity of the Al component is considered proportional to the Al concentration.

[0024] When a gradient is present in the Al concentration in the Al-augmented layer, the ratio CB/C_0 (hereinafter, may be called gradient) is preferably not greater than 0.9.

[0025] The above-mentioned ceramic heater having the Al-augmented layer can be manufactured by the following method of the present invention. The method comprises the step of firing a green body or a calcined body of the silicon nitride ceramic substrate while an Al component source substance is in contact with the surface of the green body or calcined body, to thereby form in a surface layer portion of the obtained silicon nitride ceramic substrate an Al-augmented layer having an Al concentration higher than that of an internal portion of the silicon nitride ceramic substrate. The method has the following advantages. The Al component is diffused from the surface of the substrate to thereby easily form the Al-augmented layer. Firing and formation of the Al-augmented layer can be carried out in a single process, thereby enhancing work efficiency. The Al concentration of the Al-augmented layer can be easily adjusted. Through unidirectional diffusion of the Al component, a gradient can be easily introduced in the Al concentration of the Al-augmented layer.

[0026] The above-mentioned firing process may comprise the steps of: forming a coating layer containing the Al component source substance on the surface of a cavity of a pressing die; and hot pressing the green body or the

calcined body by use of the pressing die. A parting component may be contained in the coating layer containing the Al component source substance. In this case, in a process for applying a parting material to the surface of the pressing die, the Al component source substance can be simultaneously applied to the die surface, thereby shortening the manufacturing process.

[0027] In order to contain the Al component in the ceramic substrate in the form of an inorganic compound other than Al_2O_3 as mentioned previously, firing can be performed at a temperature not lower than 1700°C in a nitrogen atmosphere having a partial pressure of oxygen (contained, for example, as an impurity) of 0.01 Pa to 100 Pa and a partial pressure of nitrogen not lower than approximately 5×10^4 Pa (approx. 0.5 atmosphere).

[0028] Embodiments of the invention will now be described, by way of example only, with reference to the accompanying drawings, in which:

FIG. 1 is a partially sectional front view showing a glow plug employing a ceramic heater of the present invention; FIG. 2 is a sectional front view showing the ceramic heater;

FIGS. 3A and 3B are views used for explaining a step of a process of manufacturing the ceramic heater, wherein FIG. 3A is a sectional view showing injection molding performed for manufacturing a heating member, and FIG. 3B is a schematic view of a unitary green body of the heating member;

FIGS. 4A and 4B are views used for explaining a subsequent step of the process of manufacturing the ceramic heater, wherein FIG. 4A is an exploded perspective view of a composite green body of the ceramic heater, and FIG. 4B is a sectional view of the composite green body;

FIGS. 5A and 5B are views used for explaining a subsequent step of the process of manufacturing the ceramic heater, wherein FIG. 5A is a schematic sectional view showing formation of the composite green body through pressing, and FIG. 5B is a schematic sectional view showing hot-pressing of the composite green body;

FIG. 6 is a schematic view showing a change in the shape of a cross section while the composite green body changes to a sintered body in the course of firing;

FIG. 7 is a sectional view showing another embodiment of the ceramic heater of the present invention;

FIGS. 8A to 8C are process-explaining diagrams which show the characteristic portion of an embodiment of the method of manufacturing the ceramic heater according to the present invention, and a modification thereof;

FIG. 9 is a process-explaining diagram showing a method of forming on a calcined body a coating layer containing Al source powder;

FIGS. 10A to 10D are views used for explaining actions in the embodiment of the method of manufacturing the ceramic heater according to the present invention;

FIG. 11 is a graph showing definition of the thickness of an Al-augmented layer;

FIG. 12 is a schematic view showing an example of the structure of the silicon nitride ceramic substrate of the ceramic heater of the present invention;

FIGS. 13A is a partial photograph of the cross section of a sample which was manufactured in a test example and falls within the scope of the present invention, showing a secondary mapping image obtained by EPMA with respect to Al;

FIG. 13B is a partial photograph of the cross section of the sample, showing a secondary mapping image obtained by EPMA with respect to O;

FIG. 14A is a sketch of the cross section of the sample obtained through observation by means of SEM, showing the micro structure of an internal portion of the sample;

FIG. 14B is a sketch of the cross section of the sample obtained through observation by means of SEM, showing the microstructure of an surface layer portion of the sample; and

FIG. 15 is a profile obtained through X-ray diffractometry of the surface layer portion of the sample.

[0029] FIG. 1 shows a glow plug using a ceramic heater manufactured by a method of the present invention while an internal structure thereof is partially exposed. A glow plug 50 includes a ceramic heater 1 disposed at one end thereof; a metal sleeve 3 which covers the ceramic heater 1 such that an end portion 2 of the ceramic heater 1 projects from the metal sleeve 3; and a cylindrical metal housing 4 which covers the sleeve 3. The ceramic heater 1 and the sleeve 3 are brazed together, whereas the sleeve 3 and the metal housing 4 are brazed together.

[0030] One end portion of a connection member 5 is fitted to a rear end portion of the ceramic heater 1. The connection member 5 is formed of a metal wire such that opposite end portions thereof are each formed into the shape of a helical spring. The other end portion of the connection member 5 is fitted to a corresponding end portion of a metal shaft 6 inserted into the metal housing 4. A rear portion of the metal shaft 6 extends to the exterior of the metal housing 4. A nut 7 is engaged with a thread portion 6a, which is formed on the peripheral surface of the extended portion of the metal shaft 6, and is tightened toward the metal housing 4 to thereby fixedly attach the metal shaft 6 to the metal housing 4. An insulation bush 8 is interposed between the nut 7 and the metal housing 4. An external thread portion 5a is formed on the external surface of the metal housing 4 and is adapted to fixedly attach the glow plug 50 to an

unillustrated engine block.

[0031] As shown in FIG. 2, the ceramic heater 1 includes a U-shaped ceramic resistance-heating member (hereinafter, called merely a heating member) 10. End portions of wire-like or rod-like electrodes 11 and 12 are embedded in corresponding end portions of the heating member 10. The heating member 10 and the electrodes 11 and 12 are entirely embedded in a rod-like silicon nitride ceramic substrate 13 having a circular cross section. The heating member 10 is disposed such that a direction-changing portion 10a is located at a tip end portion of the ceramic substrate 13.

[0032] Material for the ceramic substrate 13 is prepared, for example, by adding sintering aid powder, such as Er_2O_3 powder or Yb_2O_3 powder, to Si_3N_4 powder in an amount of 3% to 15% by weight. The resulting mixture is formed into a green body, which is then sintered to obtain the ceramic substrate 13. As shown in FIG. 10D, an Al-augmented layer 2s is formed in a surface layer portion of the ceramic substrate 13. The Al-augmented layer 2s has an Al concentration higher than that of an internal portion of the ceramic substrate 13. In the Al-augmented layer 2s, the Al concentration changes with a gradient; i.e., gradually increases toward the surface of the ceramic substrate 13.

[0033] Material for the heating member 10 is prepared, for example, by the steps of mixing Si_3N_4 powder and WC or MoSi_2 powder, which is conductive ceramic powder; and adding to the mixture sintering aid powder similar to that used for the ceramic substrate 13, in an amount of 0.8% by weight to 10.5% by weight. The resulting mixture is formed into a green body, which is then sintered to obtain the heating member 10. The sintered body has a microstructure that WC or MoSi_2 grains are dispersed in an Si_3N_4 main phase. The electrodes 11 and 12 are made of a metal wire of, for example, W, W-Re, Mo, Pt, Nb, Ta, or Nichrome.

[0034] In FIG. 2, a thin metal layer of, for example, nickel (not shown) is formed, by a predetermined method (for example, plating or vapor deposition process), on the surface of the ceramic substrate 13 in a region including an exposed portion 12a of the electrode 12. The sleeve 3 is brazed to the thin metal layer to thereby be fixedly attached to the ceramic substrate 13 and electrically connected to the electrode 12. Similarly, a thin metal layer is formed on the surface of the ceramic substrate 13 in a region including an exposed portion 11a of the electrode 11. The connection member 5 is brazed to the thin metal layer. Thus, power is supplied from an unillustrated power supply to the heating member 10 via the metal shaft 6 (FIG. 1), the connection member 5, and the electrode 11. The heating member 10 is grounded via the electrode 12, the sleeve 3, the metal housing 4 (FIG. 1), and an unillustrated engine block.

[0035] A method for manufacturing the ceramic heater 1 will next be described. First, as shown FIG. 3A, electrode materials 30 are disposed in a mold 31 having a U-shaped cavity 32, which corresponds to the heating member 10, such that respective end portions are inserted into the cavity 32. A compound 33 is injected into the cavity 32. The compound 33 contains a binder (an organic binding agent) and a material ceramic powder composed of sintering aid powder and a predominant amount of WC or MoSi_2 powder and Si_3N_4 powder. The electrode materials 30 and a U-shaped heating-member green body 34 are integrated into a unitary green body 35 shown in FIG. 3B. The heating-member green body 34 is formed such that the cross section thereof assumes a substantially circular shape.

[0036] Material powder for the ceramic substrate 13 is die-pressed into half green bodies 36 and 37 shown in FIG. 4A. A depression 38 having a shape corresponding to the unitary green body 35 is formed on a mating surface 39a of each of the half green bodies 36 and 37. Next, the half green bodies 36 and 37 are joined together at the mating surfaces 39a, while the unitary green body 35 is accommodated in the depressions 38. Then, as shown in FIG. 5A, an assembly of the half green bodies 36 and 37 and the unitary green body 35 is placed in a cavity 61a of a die 61 and is then pressed by means of punches 62 and 63, thereby obtaining a composite green body 39 as shown in FIGS. 5B and 6A. The pressing direction is substantially perpendicular to the mating surfaces 39a of the half green bodies 36 and 37.

[0037] In order to remove a binder component and the like contained in the material powder, the thus-obtained composite green body 39 is calcined at a predetermined temperature (for example, approximately 600°C) to thereby become a calcined body 39' (a calcined body can be considered a composite green body in the broad sense) shown in view (b) of FIG. 6. Subsequently, as shown in FIG. 5B, the calcined body 39' is placed in cavities 65a of hot-pressing dies 65 made of graphite or the like.

[0038] A parting material containing the Al component source substance is applied to the surface of the cavity 65a. For example, as shown in FIG. 8A, parting material powder 70 (for example, fine powder of boron nitride (BN)) and Al source powder 71 (for example, alumina powder), together with a dispersant, are placed in a solvent (for example, ethanol) to thereby prepare a coating suspension SL. The coating suspension SL is manually applied to the cavity surface by means of a brush 80 as shown in FIG. 8B or sprayed on the cavity surface by means of a spray nozzle 81 as shown in FIG. 8C. Subsequently, the solvent is allowed to evaporate for drying, thereby forming a composite coating layer 72 which contains the parting material powder 70 and the Al source powder 71.

[0039] For uniform coating and in order to reliably obtain a required coating layer thickness, the coating suspension SL contains solid matter, which is composed predominantly of the parting material powder 70 and the Al source powder 71, in an amount of preferably 3% to 20% by volume. In order to provide favorable parting performance and reliably obtain an appropriate thickness of an Al-augmented layer to be formed, the composite coating layer 72 has a thickness of preferably 25 μm to 100 μm . Similarly, the percentage of the content of the Al source powder 71 to the total content

of the parting material powder 70 and the Al source powder 71 is preferably 10% to 80% by weight. In order to uniformly diffuse the Al component during sintering, the Al source powder 71 has an average grain size of preferably 0.5 μm to 10 μm .

[0040] As shown in FIG. 5B, the calcined body 39' is placed in a firing furnace (hereinafter referred to as "furnace") 64 while being held between the pressing dies 65, which are coated with the composite coating layer 72 as mentioned above. In the furnace 64, the calcined body 39' is sintered at a predetermined temperature not lower than 1700°C (for example, approximately 1800°C) while being pressed between the pressing dies 65, to thereby become a sintered body 90 as shown in view (c) of FIG. 6. The heating-member green body 34 shown in FIG. 4B is sintered into the heating member 10, and half green bodies 36 and 37 are sintered into the ceramic substrate 13. The electrode materials 30 become the electrodes 11 and 12.

[0041] As shown in FIG. 10B, the Al component diffuses into the sintered body 90 from the Al source powder 71 contained in the composite coating layer 72 shown in FIG. 10A, to thereby form an Al-augmented layer 90s in which the Al concentration increases toward the surface of the sintered body 90. Firing is performed, for example, in the following manner. Nitrogen of the atmospheric pressure which includes impurity oxygen having a partial pressure of 0.01 Pa to 100 Pa is introduced into the furnace 64. The furnace temperature is raised to a predetermined firing temperature (for example, 1800°C) and maintained at the temperature for firing.

[0042] FIG. 12 schematically shows an example of ceramic structure in the Al-augmented layer 90s. The Al component diffusing from the composite coating layer 72 replaces, for example, the silicon component in the silicon nitride main phase to thereby form an Si-Al composite nitride (or oxynitride). Also, in the grain boundary phase, the diffusing Al component forms predominantly an inorganic compound other than alumina, such as a composite nitride, composite oxide, or composite oxynitride with another sintering aid component. When a sintering aid contains a rare-earth component R, a crystalline composite oxide with silicon, such as R_2SiO_5 and/or $\text{R}_2\text{Si}_2\text{O}_7$, is formed. In any case, the Al component is present not in the form of alumina, which tends to accelerate vitrification of the grain boundary phase, but in the form of a composite nitride or a composite oxynitride which cover the surface of the main phase, or in the form of a crystalline composite inorganic compound which is dispersed in the grain boundary phase. Thus, growth of crystal grains in the main phase is suppressed; therefore, the resulting microstructure is unlikely to involve local grain growth, which tends to cause variations in strength among ceramic substrates.

[0043] In firing mentioned above, the calcined body 39' is fired while being compressed in the direction parallel with the mating surfaces 39a of the half green bodies 36 and 37 as shown in view (c) of FIG. 6, to thereby be formed into the sintered body 90. As shown in view (c) of FIG. 6, a straight portion 34b of the heating-member green body 34 is deformed such that the circular cross section thereof is squeezed in the above-mentioned direction of compression, to thereby become a straight portion 10b of the heating member 10 having an elliptic cross section. As shown in FIG. 10B, as a result of diffusion of the Al component into the sintered body 90, the composite coating layer 72 formed on the surface of the cavity 65a of the pressing die 65 becomes a parting layer 75 composed of residual parting material. As shown in FIG. 10C, by virtue of the parting layer 75, the sintered body 90 can be readily parted from the cavity 65a.

[0044] The external surface of the thus-obtained sintered body 90 is, for example, polished such that the cross section of the ceramic substrate 13 assumes a circular shape, thereby yielding the ceramic heater 1. In order to reliably obtain the Al-augmented layer 2s having a necessary and sufficient thickness even after polishing, the thickness of the Al-augmented layer 90s must be adjusted so as to include a polishing allowance.

[0045] Notably, Al source powder may be applied to the surface of a calcined body. For example, as shown in FIG. 9, the calcined body 39' is dipped into the suspension SL in which the Al source powder is suspended, to thereby coat the calcined body 39' with the Al source powder. Subsequently, the calcined body 39' is dried to thereby form a coating layer 73 of the Al source powder on the calcined body 39'. The coating layer 73 may or may not contain parting material powder. When the coating layer 73 does not contain the parting material powder, a pressing die is preferably coated with a parting material. Notably, the suspension SL may be applied by a method other than dipping; for example, by spraying.

[0046] The resistance-heating member 10 may be formed in a different manner. As shown in FIG. 7, a paste of conductive ceramic powder is pattern-printed on a green body of a ceramic substrate in the form of a heating member. Subsequently, the green body is fired to thereby form the resistance-heating member 10 therein. Alternatively, the resistance-heating member 10 may be formed of a metal of high melting point, such as W or W-Re.

EXAMPLES

[0047] Material powder for a ceramic substrate was prepared in the following manner. Various sintering aid powders were added to Si_3N_4 powder having an average grain size of 0.7 μm according to the composition shown in Table 1. The resulting mixture was wet-pulverized in a ball mill. A predetermined amount of binder was added to the resulting slurry. The resulting slurry was dried by a spray dry process, thereby yielding the material powder for a ceramic substrate. Material powder for a heating member was prepared in the following manner. 55% by mass WC powder having

an average grain size of 1 μm , 40% by mass silicon nitride (Si_3N_4) powder, and 5% by mass Er_2O_3 powder serving as a sintering aid were mixed (the WC powder, the silicon nitride powder, and the Er_2O_3 powder were mixed so as to make a total amount of 100% by mass). The resulting mixture and a solvent were wet-mixed for 50 hours in a ball mill, followed by drying. To the resulting substance, polypropylene and wax serving as organic binders were added to thereby obtain a compound, followed by pelletization. Through use of the resulting pellets, injection molding was performed as shown in FIG. 3A, thereby yielding the unitary green body 35.

[0048] Through use of the above-mentioned material powder for a ceramic substrate, the half green bodies 36 and 37 shown in FIG. 4A were formed by the previously described method. The half green bodies 36 and 37 and the unitary green body 35 were integrally pressed into the composite green body 39 shown in FIG. 5A and view (a) of FIG. 6. The composite green body 39 was calcined at approximately 600°C to thereby obtain the calcined body 39' shown in view (b) of FIG. 6, followed by firing through hot pressing. Sample No. 9, which falls within the scope of the present invention, was formed in the following manner. Alumina powder having an average grain size of 2 μm (50 parts by mass) was mixed with BN powder having an average grain size of 1 μm (50 parts by mass). The resulting mixture was suspended in ethanol to thereby obtain a coating suspension. The coating suspension was applied to the cavity surface of a hot-pressing die, followed by drying to thereby form a composite coating layer having a thickness of approximately 50 μm on the cavity surface. Firing conditions were as follows: firing temperature 1800°C; pressure 300 kgf/cm²; and firing time 30 minutes. Nitrogen of the atmospheric pressure which includes impurity oxygen having a partial pressure of 0.01 Pa to 100 Pa was introduced into a furnace and heated to a temperature of 1800°C, thereby establishing a firing atmosphere. The calcined body 39' was fired under the firing conditions in the firing atmosphere.

[0049] Sample No. 9 was cut perpendicularly to the axis thereof. The cut surface was polished and subjected to EPMA for distribution of elements. FIGS. 13A and 13B show the results. FIG. 13A shows intensity mapping of characteristic X-ray of Al (aluminum), and FIG. 13B shows that of O (oxygen). In FIGS. 13A and 13B, an arcuate contour of the cross section of Sample No. 9 appears at an upper portion of the image. In the image of the cross section, a whitish region is a region of high characteristic X-ray intensity; i.e., a region of high component concentration. As seen from FIG. 13A, a region in which the Al concentration is increased apparently; i.e., an Al-augmented layer is formed in a surface layer portion of Sample No. 9 (silicon nitride ceramic substrate). Furthermore, in Sample No. 9, the Al concentration decreases toward the internal portion. Line analysis which was conducted separately by means of EPMA revealed that the average thickness of the Al-augmented layer defined in FIG. 11 was approximately 280 μm .

[0050] FIG. 14A shows the microstructure of an internal portion of Sample No. 9 as observed through a scanning electron microscope (SEM), whereas FIG. 14B shows that of the Al-augmented layer. As seen from FIGS. 14A and 14B, the Al-augmented layer is finer in crystal grains than the internal portion. As estimated from the image of FIG. 14A, the average grain size of crystal grains (main phase) of the internal portion is approximately 0.36 μm , and the maximum grain size is approximately 5 μm . As estimated from the image of FIG. 14B, the average grain size of crystal grains (main phase) of the Al-augmented layer is approximately 0.28 μm , and the maximum grain size is approximately 2.5 μm .

[0051] If the Al component is present in the form of Al_2O_3 in the Al-augmented layer, corresponding augmentation of the oxygen component must be observed in the layer. However, as seen from comparison of FIGS. 13A and 13B, such augmentation of oxygen is not observed. The present inventors studied the detailed distribution of the oxygen concentration by line analysis which was carried out by means of EPMA, and found that the oxygen concentration of the Al-augmented layer is rather lower than that of the internal portion. This implies that the Al component is present in the form of an inorganic compound other than Al_2O_3 .

[0052] FIG. 15 shows diffraction patterns obtained by X-ray diffractometry conducted as follows: the $\text{K}\alpha 1$ ray (wavelength: approx. 1.54 angstroms) of Cu is caused to impinge on the surface of the ceramic substrate. As seen from FIG. 15, crystals of a composite oxide with Er and Si are present in addition to $\beta\text{-Si}_3\text{N}_4$, which is the main phase.

[0053] The ceramic heater samples were polished such that the ceramic substrate assumes a diameter of 3.50 mm (arithmetic average roughness of polished surface: 0.1 μm) and were then subjected to a 3-point bending test (span length: 12 mm; crosshead speed: 0.5 mm/min) at room temperature (number of samples N = 10) as well as at 1400°C (number of samples N = 5). In the case of Sample No. 9, even after polishing, the Al-augmented layer remained by an average thickness of approximately 280 μm . The test results are shown in Table 1.

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Table 1

Sample No.	Sintering Aid	Composition (mol %)				Room Temp. (N = 10)	Bending Strength (MPa) (Minimum/Average)
		SiN	RE ₂ O ₃	SiO ₂	Al ₂ O ₃		
1	Al ₂ O ₃ -Y ₂ O ₃	90	2Y ₂ O ₃		8Al ₂ O ₃	1143/1212	251/320
2	Al ₂ O ₃ -Y ₂ O ₃	90	4Y ₂ O ₃		4Al ₂ O ₃	1125/1191	270/335
3	Er ₂ O ₃ -SiO ₂	92	3Er ₂ O ₃	5SiO ₂	-	1010/1164	661/739
4	Er ₂ O ₃ -SiO ₂	90	3Er ₂ O ₃	7SiO ₂	-	982/1189	640/725
5	Er ₂ O ₃ -SiO ₂	83	5Er ₂ O ₃	7SiO ₂	-	1053/1205	647/711
6	Er ₂ O ₃ -SiO ₂ -Al ₂ O ₃	91.5	3Er ₂ O ₃	7SiO ₂	0.5Al ₂ O ₃	1083/1213	492/545
7	Er ₂ O ₃ -SiO ₂ -Al ₂ O ₃	91	3Er ₂ O ₃	7SiO ₂	1Al ₂ O ₃	1177/1241	416/481
8	Er ₂ O ₃ -SiO ₂ -Al ₂ O ₃	90	3Er ₂ O ₃	7SiO ₂	2Al ₂ O ₃	1165/1228	385/457
9	Er ₂ O ₃ -SiO ₂ + Al-augmented layer	92	3Er ₂ O ₃	7SiO ₂	Al-augmented layer	1202/1248	652/731

[0054] The test results shown in Table 1 reveal the following. Samples using an Al_2O_3 - Y_2O_3 -type sintering aid (Nos. 1 and 2) show a significant impairment in high-temperature strength as measured at 1400°C. In the case of samples using an Er_2O_3 - SiO_2 -type sintering aid (Nos. 3 to 5), the minimum room-temperature strength is low, and variation in room-temperature strength is great. In the case of samples using an Er_2O_3 - SiO_2 - Al_2O_3 -type sintering aid (Nos. 6 to 8), variation in room-temperature strength is minor, but high-temperature strength is rather low. In the case of Sample No. 9, which falls within the scope of the present invention, high-temperature strength is sufficiently high, and variation in room-temperature strength is minor.

[0055] Obviously, numerous modifications and variations of the present invention are possible in light of the above teachings. It is therefore to be understood that within the scope of the appended claims, the present invention may be practiced otherwise than as specifically described herein.

Claims

15. 1. A ceramic heater comprising a silicon nitride ceramic substrate (13), and a resistance-heating member (10) embedded in said silicon nitride ceramic substrate (13), **characterized in that**
an Al-augmented layer (2s) is formed in a surface layer portion of said silicon nitride ceramic substrate (13), and wherein the Al-augmented layer (2s) has an Al concentration higher than that of an internal portion of said silicon nitride ceramic substrate (13).
20. 2. A ceramic heater according to Claim 1, **characterized in that** the Al concentration increases toward the surface of said silicon nitride ceramic substrate (13).
25. 3. A ceramic heater according to Claim 1 or 2, **characterized in that** the thickness of the Al-augmented layer (2s) is in the range of from 50 μm to 1000 μm .
30. 4. A ceramic heater according to any one of Claims 1 to 3, **characterized in that** the Al-augmented layer (2s) assumes an average Al concentration in the range of from 0.1% to 5% by weight.
35. 5. A ceramic heater according to any one of Claims 1 to 4, **characterized in that** said silicon nitride ceramic substrate (13) contains a rare-earth component, which serves as a sintering-aid component, and the rare-earth component is present in the form of a composite oxide with silicon.
40. 6. A ceramic heater according to Claim 5, **characterized in that** the composite oxide is contained in the form of R_2SiO_5 and/or $\text{R}_2\text{Si}_2\text{O}_7$, where R is the rare-earth component.
45. 7. A ceramic heater according to any one of Claims 1 to 6, **characterized in that** the average grain size of a silicon nitride main phase in the Al-augmented layer (2s) is smaller than that of the silicon nitride main phase in the internal portion.
50. 8. A ceramic heater according to Claim 7, **characterized in that** the average grain size of the silicon nitride main phase in the Al-augmented layer (2s) is in the range of from 0.1 μm to 1 μm , and that of the silicon nitride main phase in the internal portion is in the range of from 0.2 μm to 5 μm .
55. 9. A ceramic heater according to any one of Claims 1 to 8, **characterized in that** the Al component in said silicon nitride ceramic substrate (13) is present predominantly in the form of an inorganic compound other than Al_2O_3 .
10. A ceramic heater **characterized by** comprising a silicon nitride ceramic substrate (13) containing an Al component and a resistance-heating member (10) embedded in said silicon nitride ceramic substrate (13), the Al component being present in said silicon nitride ceramic substrate (13) predominantly in the form of an inorganic compound other than Al_2O_3 .
11. A ceramic heater according to any one of Claims 1 to 10, **characterized in that** the Al content of said silicon nitride ceramic substrate (13) is in the range of from 0.1% to 5% by weight, and the rare-earth component content of said silicon nitride ceramic substrate (13) is in the range of from 3% to 15% by weight.
12. A method for manufacturing a ceramic heater comprising a silicon nitride ceramic substrate (13) and a resistance-heating member (10) embedded in the silicon nitride ceramic substrate (13), said method being **characterized by**

comprising the step of:

firing a green body or a calcined body of the silicon nitride ceramic substrate (13) while an Al component source substance is in contact with the surface of the green body or calcined body, to thereby form in a surface layer portion of the obtained silicon nitride ceramic substrate (13) an Al-augmented layer (2s) having an Al concentration higher than that of an internal portion of the silicon nitride ceramic substrate (13).

5 **13. A method for manufacturing a ceramic heater according to Claim 12, characterized by further comprising the steps of:**

10 forming a coating layer containing the Al component source substance on the surface of a cavity of a pressing die; and
hot pressing the green body or the calcined body by use of the pressing die.

15 **14. A method for manufacturing a ceramic heater according to Claim 13, characterized in that the coating layer containing the Al component source substance contains a parting component.**

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FIG. 1

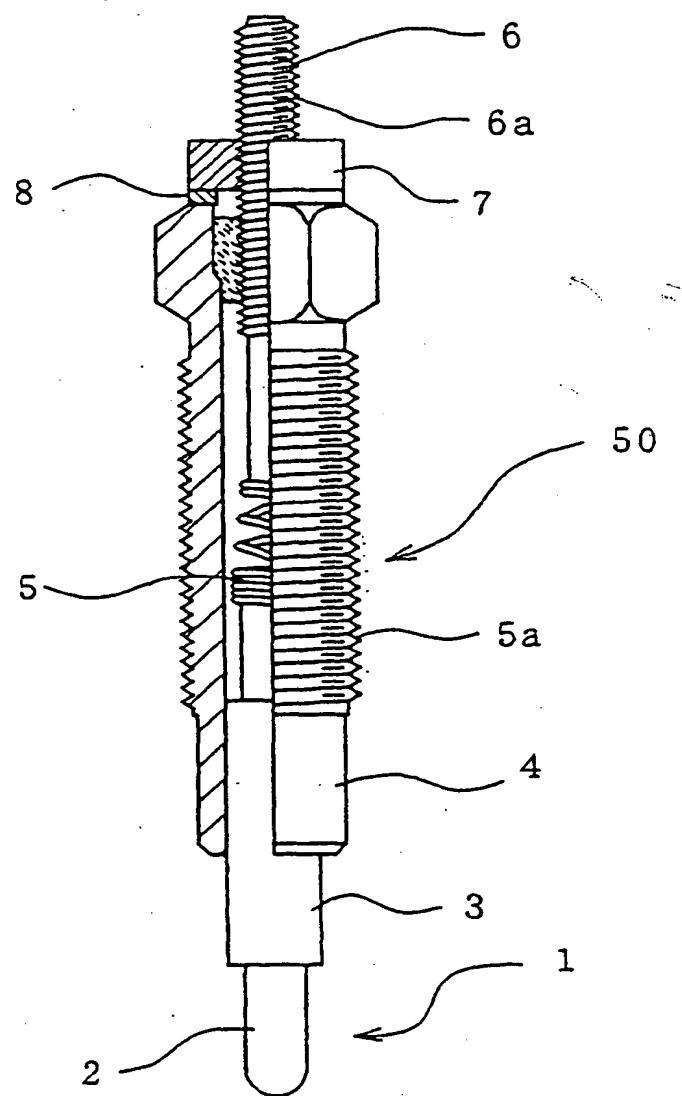


FIG. 2

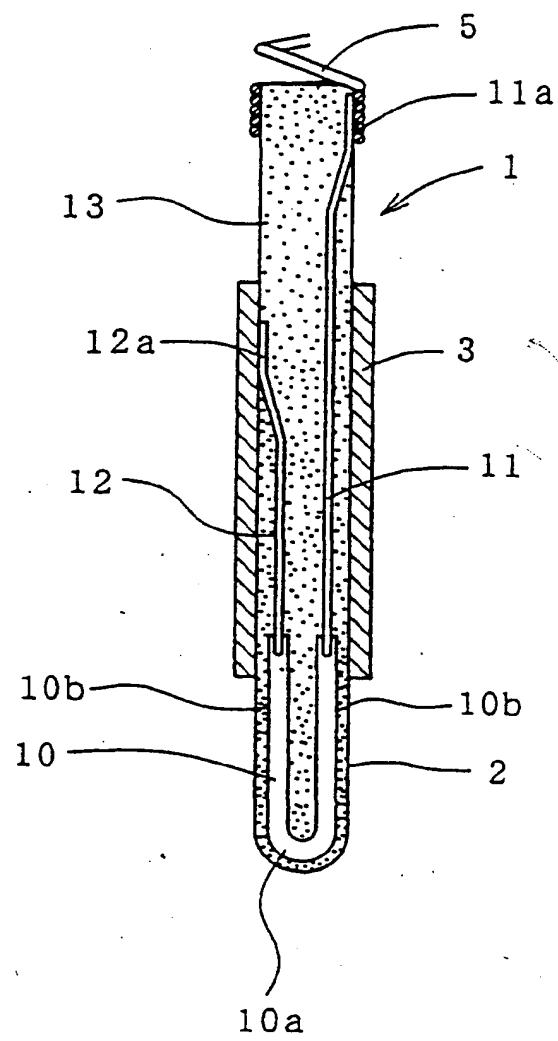


FIG. 3A

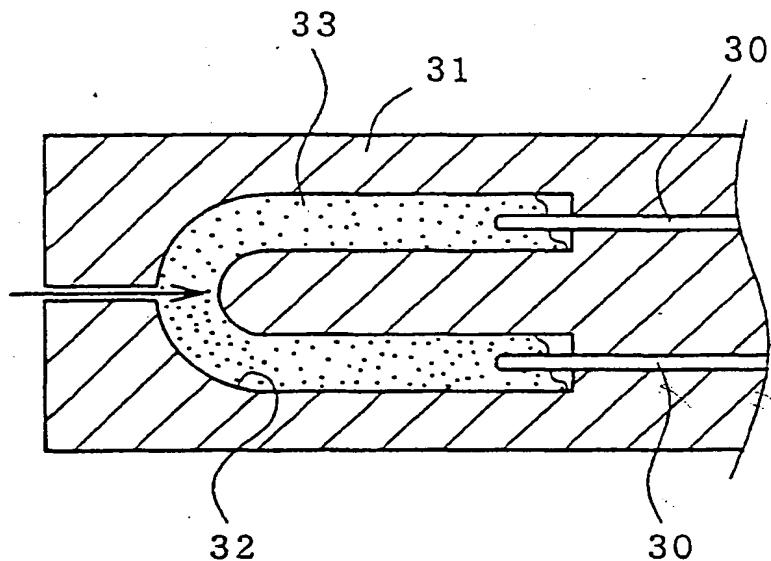


FIG. 3B

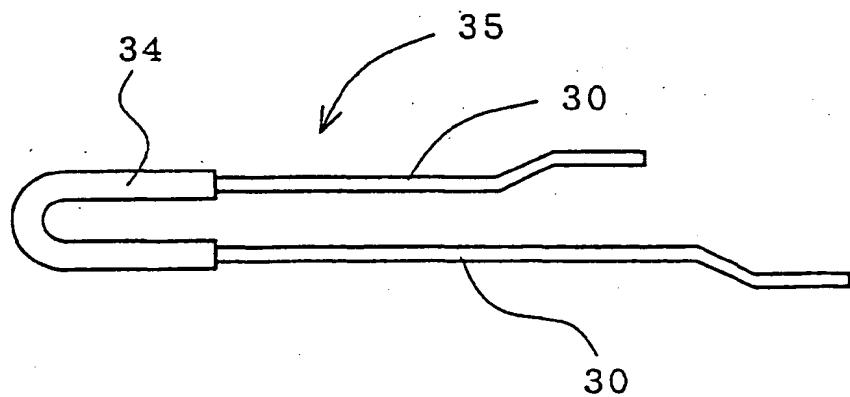


FIG. 4A

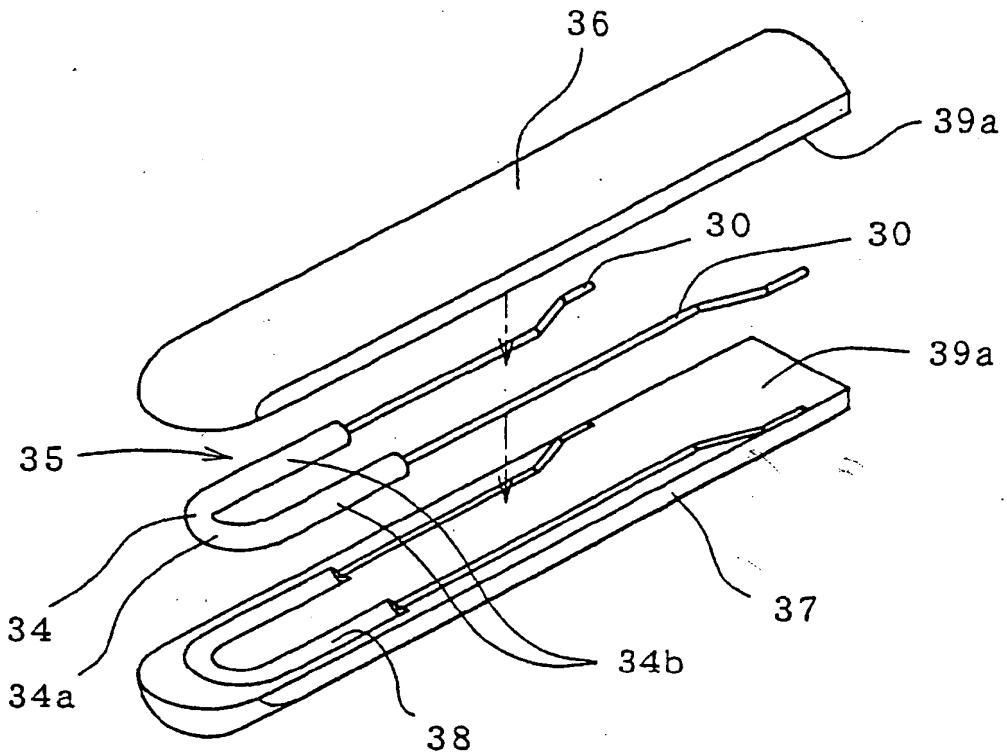


FIG. 4B

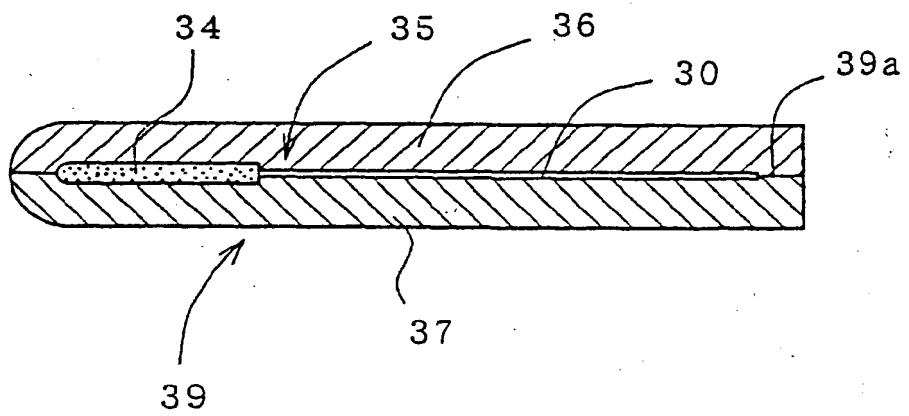


FIG. 5A

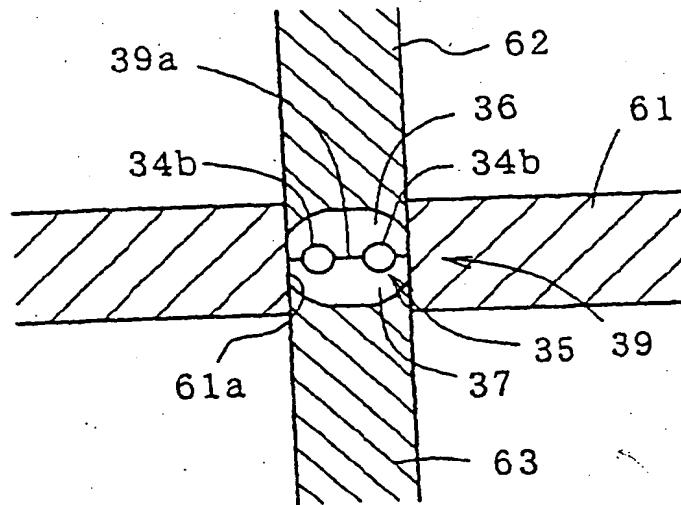


FIG. 5B

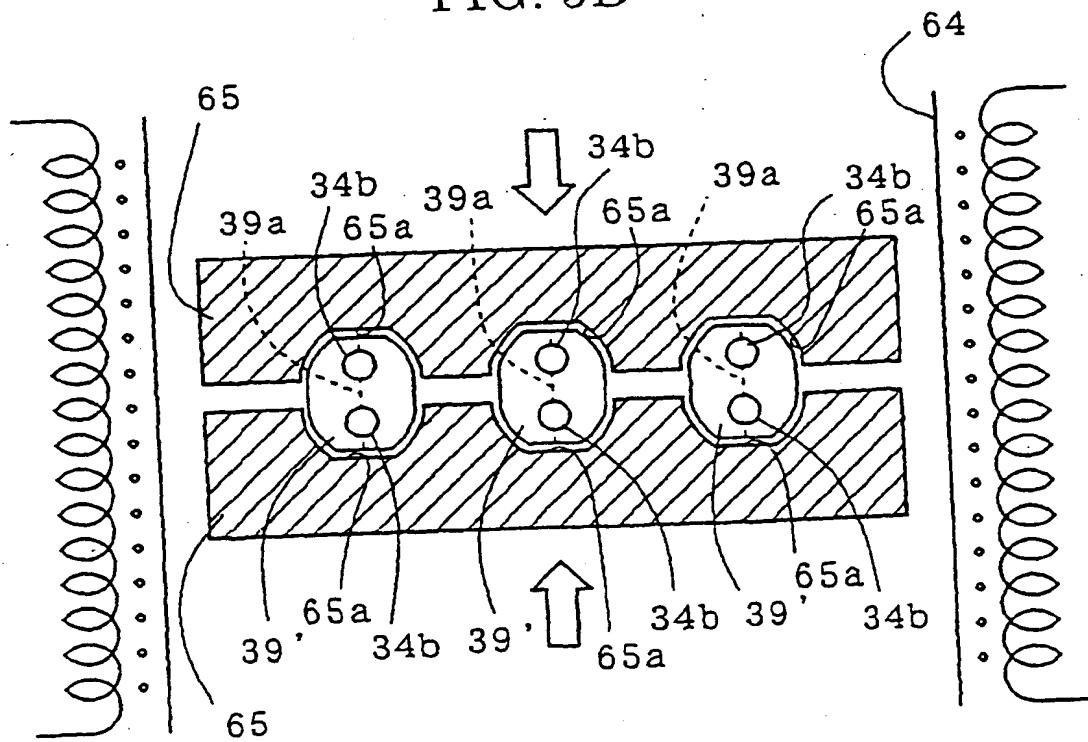


FIG. 6

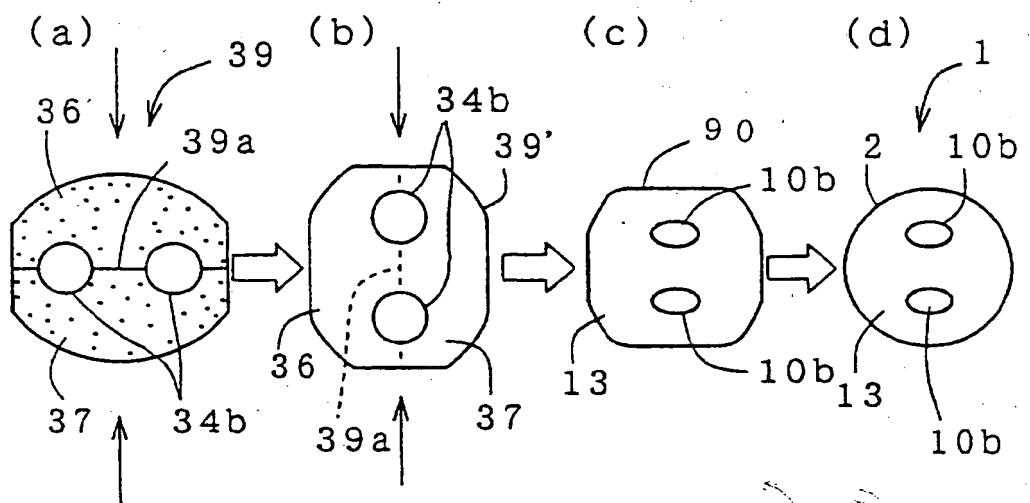


FIG. 7

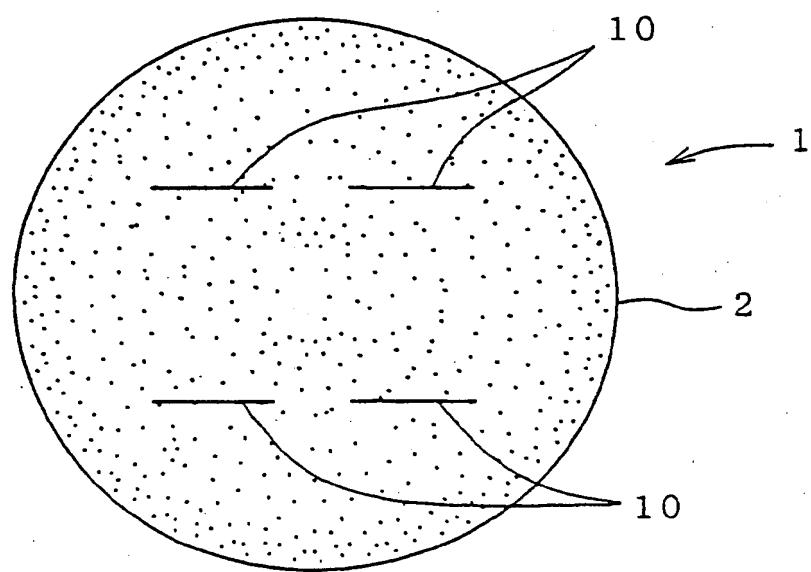


FIG. 8A

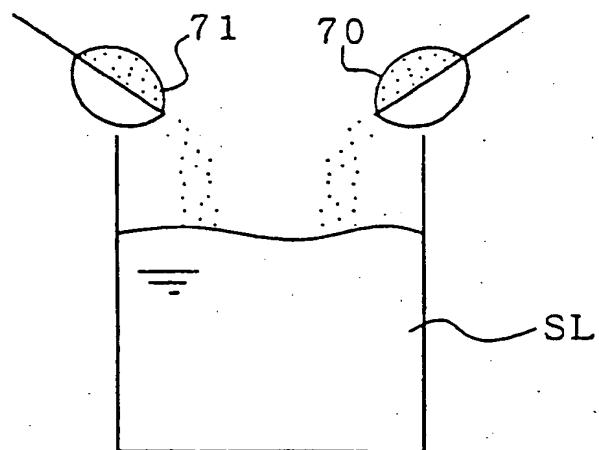


FIG. 8B

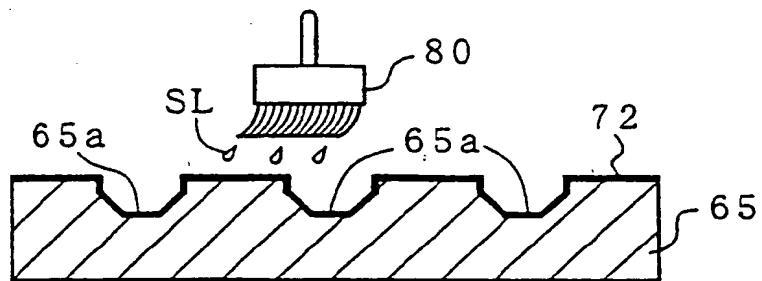


FIG. 8C

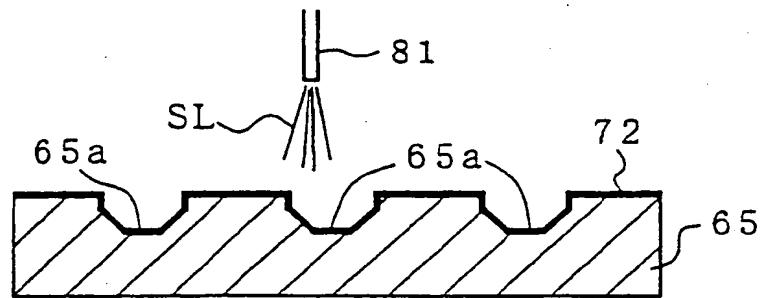


FIG. 9

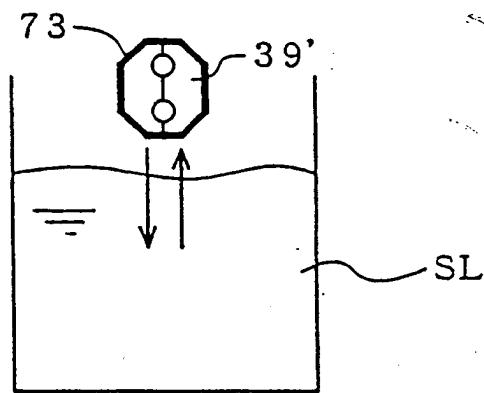


FIG. 10A

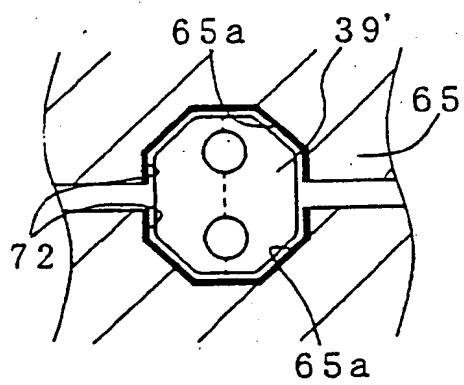


FIG. 10B

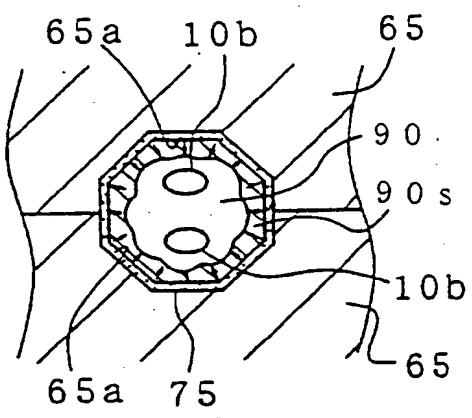


FIG. 10C

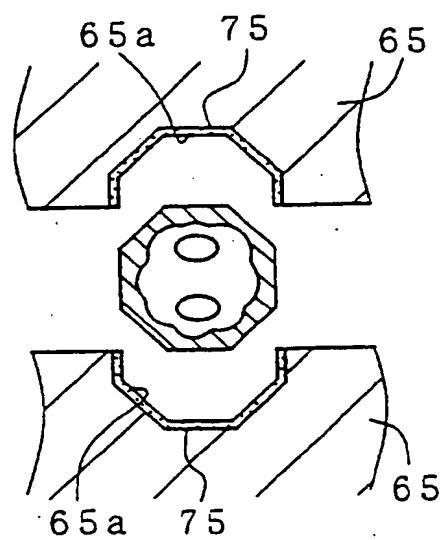


FIG. 10D

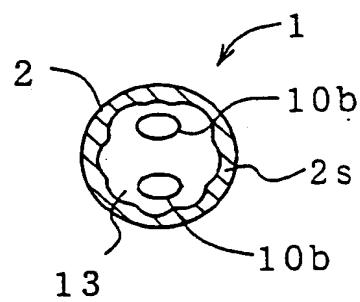


FIG. 11

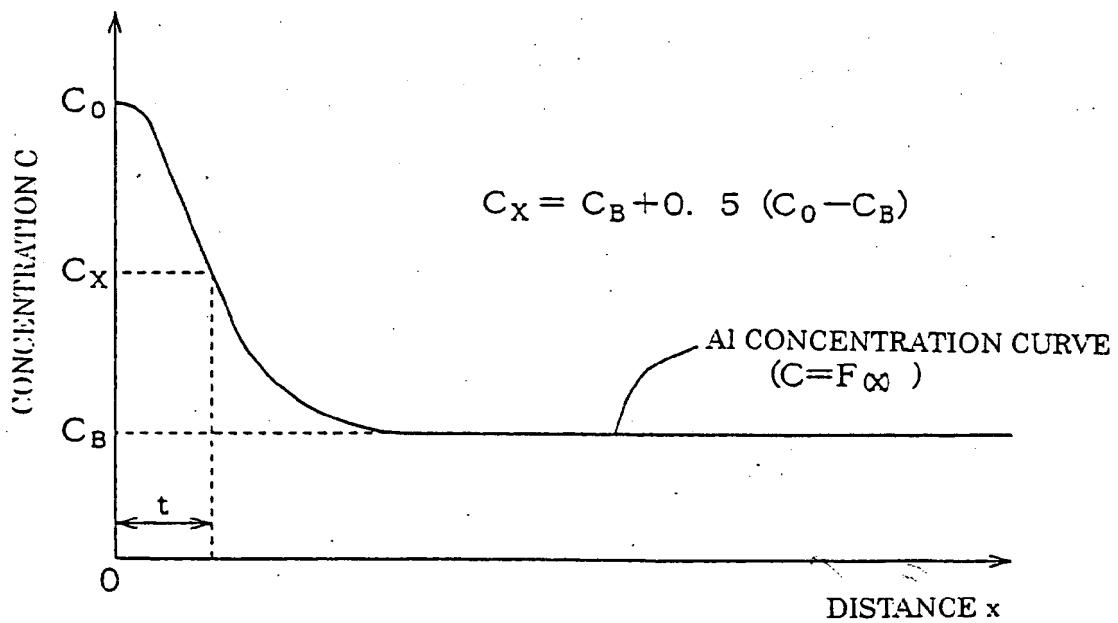


FIG. 12

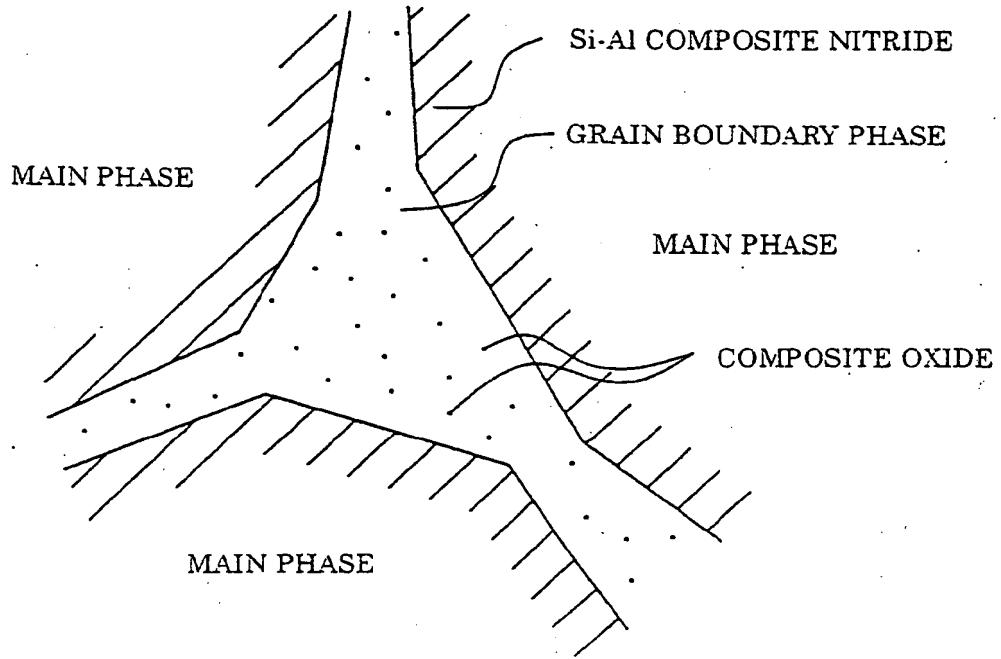


FIG. 13A

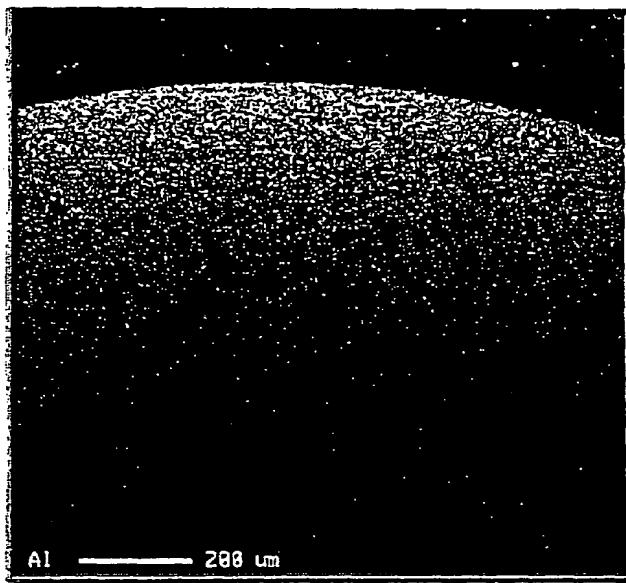


FIG. 13B

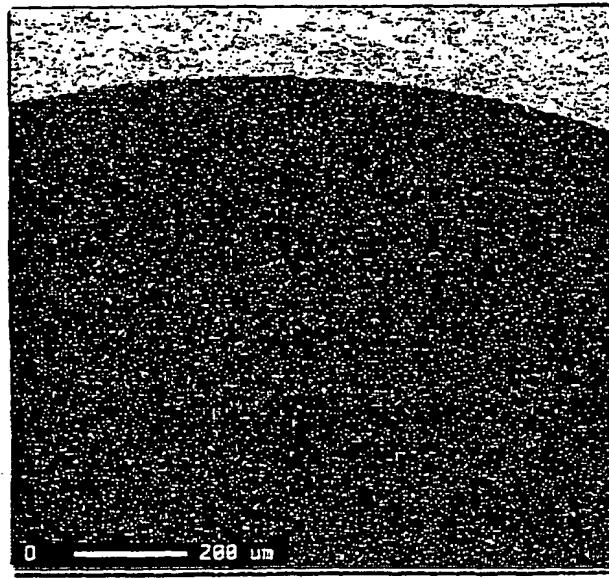
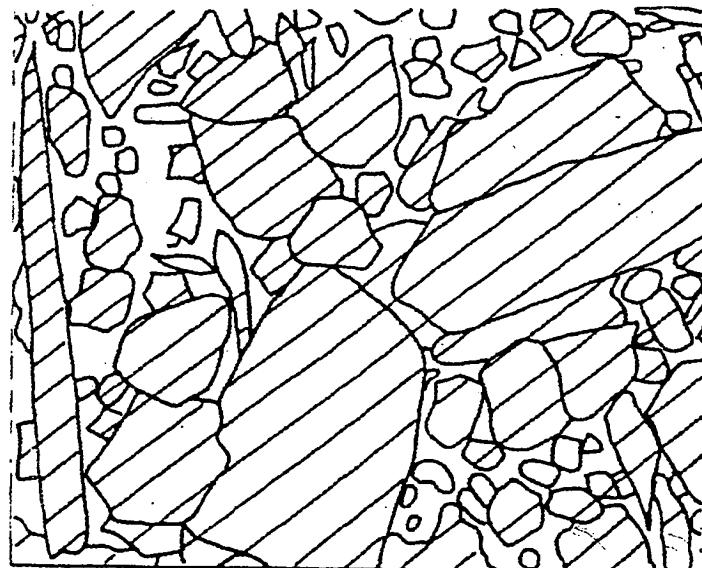
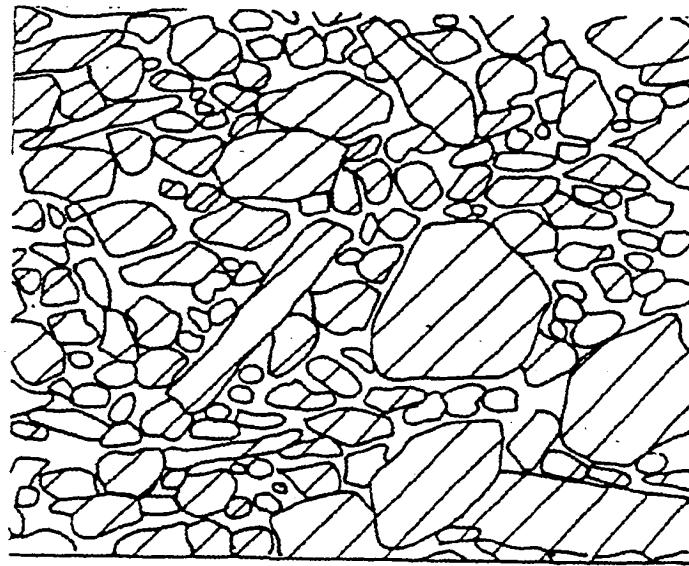


FIG. 14A



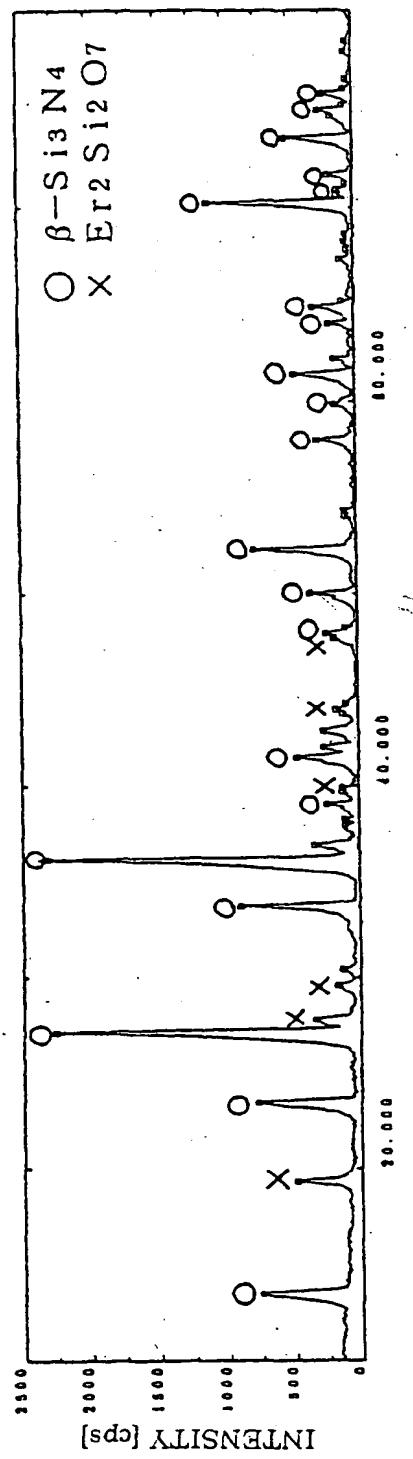
1 μm

FIG. 14 B



1 μm

FIG. 15





(19)



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(11)

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(12)

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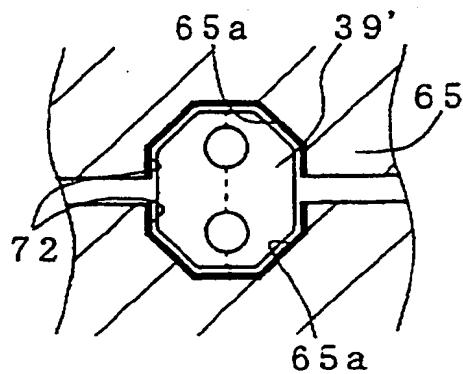
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Gray's Inn
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(54) Ceramic heater and method for manufacturing the same

(57) A ceramic heater includes a silicon nitride ceramic substrate (13) and a heating member (10) embedded in the silicon nitride ceramic substrate (13). An Al-augmented layer (2s) is formed in a surface layer portion of the silicon nitride ceramic substrate (13). The Al-aug-

mented layer (2s) has an Al concentration higher than that of an internal portion of the silicon nitride ceramic substrate (13). The Al-augmented layer (2s) assumes such a graded composition structure that the Al concentration increases toward the surface of the silicon nitride ceramic substrate (13).

FIG. 10A



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FIG. 10B

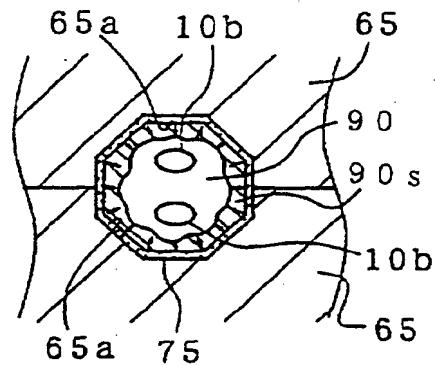


FIG. 10C

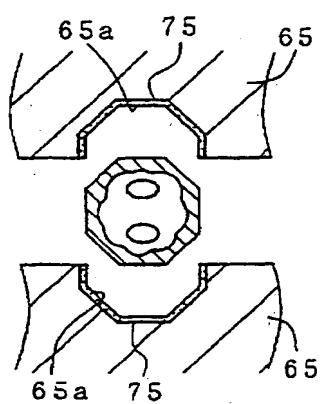
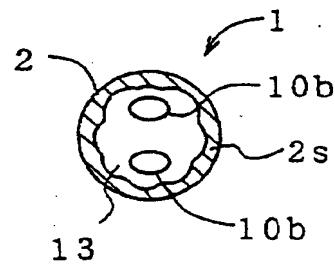


FIG. 10D





European Patent
Office

EUROPEAN SEARCH REPORT

Application Number
EP 01 30 2427

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A	US 4 929 813 A (MASAKA MITSUKE ET AL) 29 May 1990 (1990-05-29) * column 2, line 64,65 * * column 5, line 29 - line 36; figure 1 * ---	1-14							
A	US 5 750 958 A (OHNO MICHIO ET AL) 12 May 1998 (1998-05-12) * column 6, line 37 - line 46; figures 1,2,6; table 11 * ---	1-14							
A	PATENT ABSTRACTS OF JAPAN vol. 017, no. 579 (E-1451), 21 October 1993 (1993-10-21) -& JP 05 174948 A (KYOCERA CORP), 13 July 1993 (1993-07-13) * abstract * -----	1-14							
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			H05B F23Q						
<p>The present search report has been drawn up for all claims</p> <table border="1" style="width: 100%; border-collapse: collapse;"> <tr> <td style="width: 33%;">Place of search</td> <td style="width: 33%;">Date of completion of the search</td> <td style="width: 34%;">Examiner</td> </tr> <tr> <td>MUNICH</td> <td>21 November 2002</td> <td>Gea Haupt, M</td> </tr> </table> <p>CATEGORY OF CITED DOCUMENTS</p> <p>X : particularly relevant if taken alone Y : particularly relevant if combined with another document of the same category A : technological background D : non-written disclosure P : intermediate document</p> <p>T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons & : member of the same patent family, corresponding document</p>				Place of search	Date of completion of the search	Examiner	MUNICH	21 November 2002	Gea Haupt, M
Place of search	Date of completion of the search	Examiner							
MUNICH	21 November 2002	Gea Haupt, M							

**ANNEX TO THE EUROPEAN SEARCH REPORT
ON EUROPEAN PATENT APPLICATION NO.**

EP 01 30 2427

This annex lists the patent family members relating to the patent documents cited in the above-mentioned European search report. The members are as contained in the European Patent Office EDP file on The European Patent Office is in no way liable for these particulars which are merely given for the purpose of information.

21-11-2002

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FORM P0459

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